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## Regular Article

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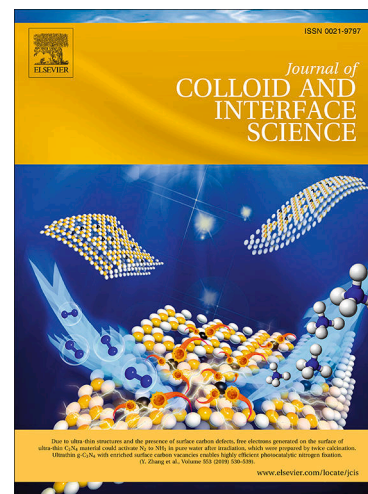
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# **Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@Graphene Oxide van der Waals Heterostructures with Enhanced Photocatalytic Activity toward Oxygen Generation**

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**ABSTRACT**

The van der Waals (vdW) integration enables to create heterostructures with intimate contact and bring new opportunities. However, it is not confined to layered materials but can also be generally extended to 3D materials. Multidimensional Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@graphene oxide (GO) van der Waals heterostructures are synthesized by one-pot wet chemistry method. Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub> composite nanoparticles are self-assembled with GO framework by vdW interaction to form vdW heterostructures, in which GO framework allows short electron transport distance and rapid charge transfer and provides massive reactive sites. Such self-assembled heterostructures show a superior high photoactivity towards oxygen evolution with an enhanced oxygen generation rate of 1828  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , nearly 3 times than that of pure BiVO<sub>4</sub>, attributed to the accelerated charge separation and transfer processes of Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO vdW heterostructures. This study indicates that our strategy provides a new avenue towards fabricating multi-dimensional vdW heterostructures and inspiring more innovative insights in oxygen evolution field.

**Keywords:** BiVO<sub>4</sub>, Graphene oxide, Van der Waals, Heterostructures, Oxygen evolution.

## 1. Introduction

Two-dimensional (2D) materials, including MoS<sub>2</sub> [1], graphite [2], WS<sub>2</sub> [3], MoSe<sub>2</sub> [4], have attracted worldwide attentions in the application in energy storage and conversion. Particularly, 2D graphene oxide (GO) has high specific surface, high strength, high stability, and high carrier mobility and shows great potentials in photocatalytic, thermal, and electrical fields [5]. Meanwhile, oxygen-rich functional groups of GO provide abundant reactive sites, which can enhance the interaction between their photocatalytic hybrid materials to facilitate electron transfer to enhance photocatalytic efficiency [6].

Currently, several universal strategies are used to assemble different 2D building blocks to form heterostructures by using physical or chemical interactions [7, 8]. Strong covalent bonds provide in-plane stability, whereas relatively weak, van-der-Waals-like forces are sufficient to keep the stack together. Therefore, these different 2D crystals can be stacked each another by using weak van der Waals (vdW) forces [9]. In fact, a few layered materials, such as Bi<sub>2</sub>Te<sub>3</sub>/FeTe [10], MoS<sub>2</sub>/Au [11] and InSe/graphene heterostructures [12, 13] have been reported to link together by vdW forces for designing heterostructures. Atomic interfaces between layered materials accelerate the vdW interaction. Therefore, the vdW integration enables to create a great deal of heterostructures with intimate contact and bring new opportunities. Multilayer graphene vdW heterostructures were constructed using hexagonal boron nitride (h-BN) as a substrate [14, 15]. Similarly, graphene/MoS<sub>2</sub> [16] and BN/graphene/BN [17] were synthesized to improve charge mobility of graphene with the construction of 2D/2D

vdW integration. Such vdW integration can promote the separation of interfacial carriers and enable access to fast charge speed.

Actually, vdW interactions are not confined to interactions in layered materials and can also be generally extended to 3D materials [18]. Various multi-dimensional (2D/3D) heterostructures with multi-functionalities can be elaborately designed by using the different rational hybridization methods of different dimensional materials [19-21]. Their properties have been significantly enhanced by compensating individual weakness. Recently, a 2D/3D vdW heterostructure consisting of 2D triazine-based framework and 3D triazine-based graphdiyne has been reported for hydrogen evolution reaction [22]. A 3D/2D/2D structured  $\text{BiVO}_4/\text{FeVO}_4@\text{rGO}$  has been reported to have an enhanced transport and separation efficiency of photogenerated carriers [23]. Likewise, 2D/3D/2D  $\text{rGO}/\text{Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$  nanocomposites were designed to inhibit the recombination rate of photoexcited charge carriers [24]. These fabricated multi-dimensional architectures, assembled with micrometer- and nanometer-scaled building blocks, exhibit unique photocatalytic activity compared to the individual structures [25, 26]. Therefore, such an approach provides various material choices and selectable properties and can be applicable for developing 3D systems.

In this work, we report a facile self-assembly of a multi-dimensional (2D/3D)  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  vdW heterostructures via one-pot “wet” chemistry approach. The synthesized  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$  particle with a small diameter shows short electron transport distance. By stacking GO framework on  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$ , 2D/3D heterostructures is formed and inherits the excellent properties of GO framework, including high specific surface

area and high carrier mobility. Such 2D/3D vdW heterostructures possess massive reactive sites and accelerated electron transfer, thus have significantly enhanced photocatalytic activity toward oxygen generation.

## 2. Experiments

### 2.1 Synthesis of 2D/3D Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO vdW heterostructures

All chemicals were obtained and used from commercial sources as analytical pure reagents without further purification. The 2D/3D Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO vdW heterostructures were self-assembled by one-pot “wet” chemistry approach. 0.4 mmol Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O (Sinopharm Chemical Reagent Co., Ltd., 99 %) was firstly dissolved in 16 mL of glycerol after stirring for 1 hour (h). Then, 0.4 mmol NaVO<sub>3</sub>·2H<sub>2</sub>O (Sinopharm Chemical Reagent Co., Ltd., 99 %) was dissolved in another 16 mL of deionized water. It can be clearly seen that it gradually transformed into transparent and homogeneous after stirring for 0.5 h. The above solutions were mixed under continuous stirring, forming an orange suspension. The mixture with GO solution was then transferred into a Teflon-lined stainless steel autoclave and then heated to 180 °C. Then, the as-synthesized nanoparticles were self-assembled by a facile solvothermal method to establish a vdW heterojunction of Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO nanocomposites. For comparison, pure BiVO<sub>4</sub> was synthesized under the same conditions by hydrothermal method without the addition of GO solution. The experiment result shows that Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO can be obtained after adding graphene oxide, indicating the GO is favorable for the formation of Bi<sub>2</sub>O<sub>3</sub>. The preparation approach can promote to fabricate mixed-dimensional vdW heterostructures in “wet” chemistry field.

### 2.2 Sample Characterization

The samples were detected by X-ray diffraction (XRD, D/Max-2200PC) and field-emission scanning electron microscope (SEM, Hitachi S4600). The transmission electron microscopy (TEM, JEM-2100F) was taken at current density of 10  $\mu\text{A}$  with an accelerating voltage of 200 kV. Raman spectra were measured on Renishaw-invia Raman spectroscopy. Ultraviolet-visible diffuse reflectance spectra (UV-Vis DRS) were characterized on a Cary 5000 spectrophotometer equipped with an integrating sphere, with  $\text{BaSO}_4$  as a reflectance standard.

### 2.3 Photocatalytic Oxygen Evolution

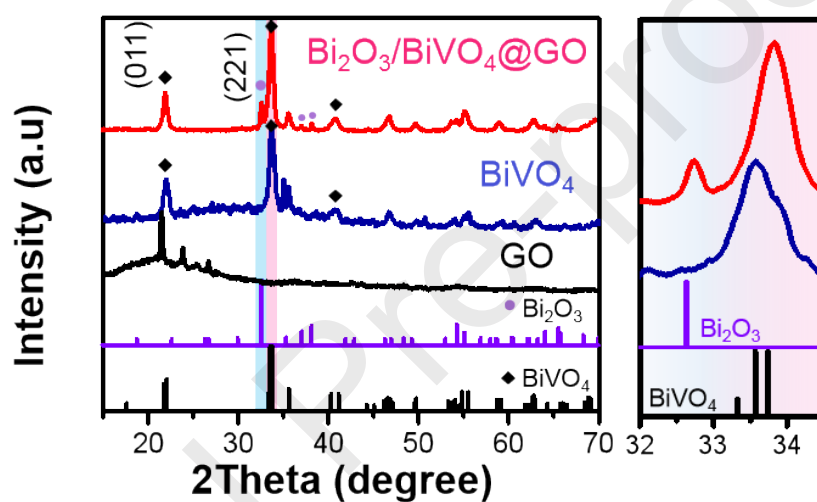
Photocatalytic  $\text{O}_2$  production is carried out under a 500 W Xe lamp. In the suspension, 50 mg catalysts were dispersed in 50 mL aqueous solution and  $\text{Fe}(\text{NO}_3)_3$  was employed as the sacrifice agent. During the  $\text{O}_2$  evolution, the solution was stirred and treated with nitrogen gas several times to remove the dissolved gases in the liquid. The experiment was conducted at room temperature by circulating cold water. After being irradiated for a period of time, the production of  $\text{O}_2$  was analyzed using gas chromatograph (9790II, Fuli, TCD) and calculated on the basis of  $\text{O}_2$  peak area.

## 3. Results and discussion

### 3.1 Characterization of photocatalyst

**Fig. 1** shows the typical X-ray diffractometry (XRD) patterns of as-prepared pure  $\text{BiVO}_4$  and 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures. As can be seen, all the products have monoclinic  $\text{BiVO}_4$ . The intensive (221) peak (inset of **Fig. 1**) shows that the  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures have  $\text{Bi}_2\text{O}_3$  phase, indicating coexistence

of  $\text{Bi}_2\text{O}_3$  and  $\text{BiVO}_4$  in the heterostructure photocatalysts. According to the results of whole pattern fitting and Rietveld refinement, the molar ratio of  $\text{Bi}_2\text{O}_3$  and  $\text{BiVO}_4$  is 4:96. In the synthesis step, GO solution enables the phase conversion and causes another product  $\text{Bi}_2\text{O}_3$ . It is worth noting that the modification of GO has no obvious influence on the XRD peaks of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$ , indicating that the vdW interaction between GO and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$  does not alter the crystal structure of samples.

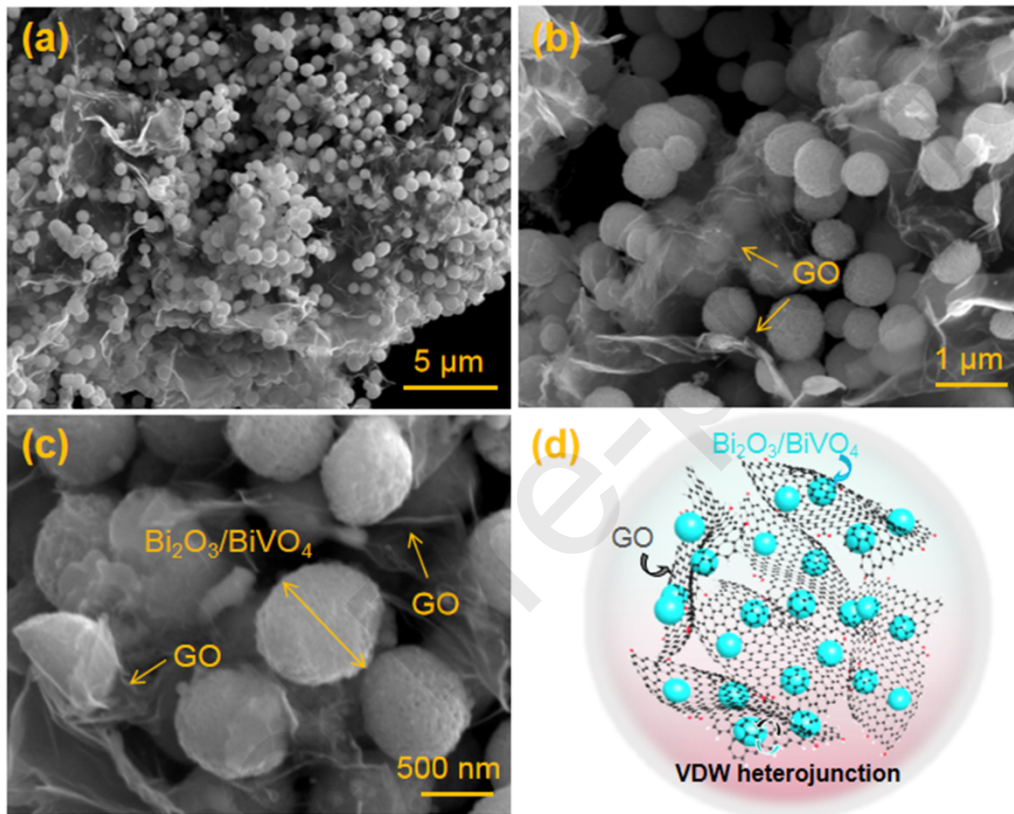


**Fig. 1.** XRD patterns of GO,  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@GO$ .

The microstructure and morphology of 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@GO$  vdW heterostructures synthesized by one-pot process described above were investigated by using SEM and TEM observations. **Fig. 2a-c** are typical SEM images of the products, which clearly reveals that the 2D/3D structure and the close integration between these three materials. As can be seen, mulberry-shape  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$  nanospheres have 500~600 nm and surrounded by GO layers to form heterostructures. Here, the GO layers act as the skeleton to enable the formation



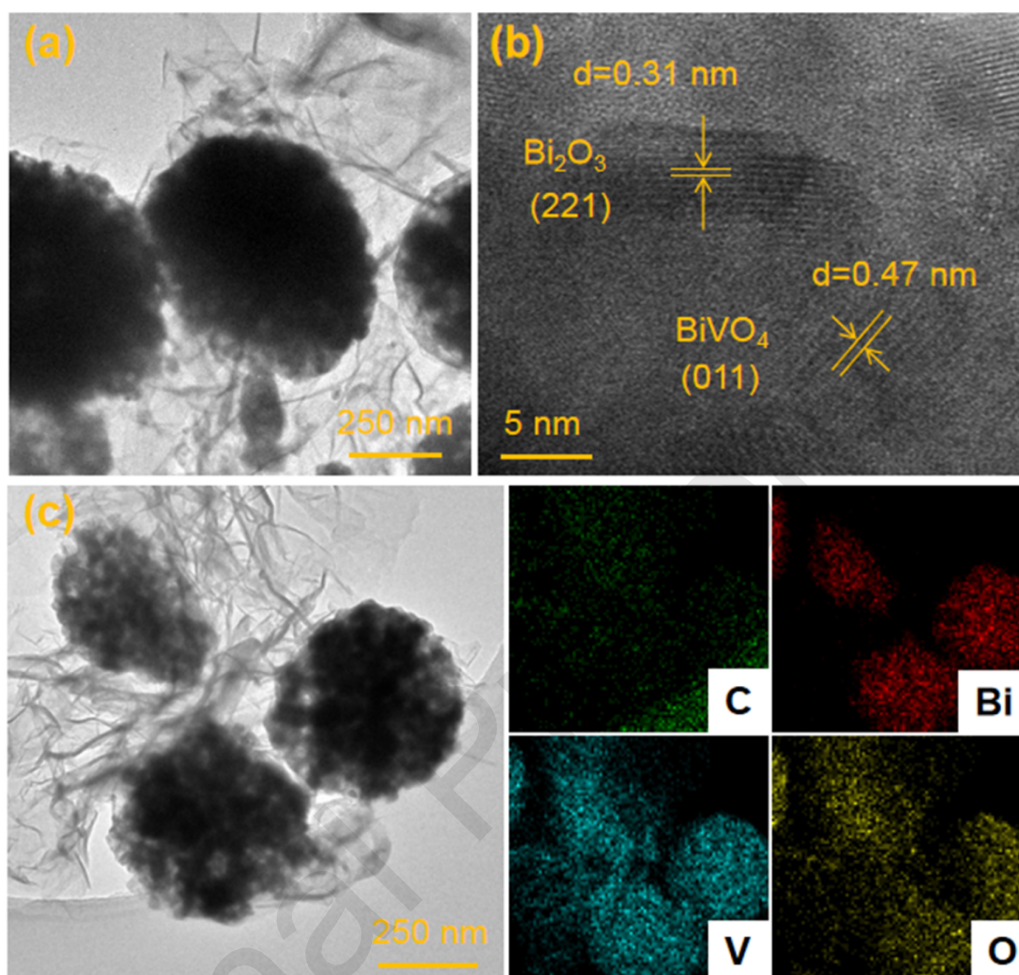
of 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures. In the nucleation stage during the hydrothermal process, a large number of nuclei form rapidly and oriented by the van der Waals interactions on GO to establish vdW heterojunctions of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ , as schematically illustrated in **Fig. 2d**.



**Fig. 2.** (a-c) SEM images and (d) schematic illustration of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ .

**Fig. 3a** is a typical low-magnification TEM images of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  vdW heterostructures and shows that the 2D GO layers surround  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$  nanospheres by a self-assembly method. **Fig. 3b** is its corresponding high-resolution TEM (HRTEM) image, in which the crystal lattice distances of 0.47 nm and 0.31 nm can be measured, corresponding to the (011) plane of  $\text{BiVO}_4$  and (221) plane of  $\text{Bi}_2\text{O}_3$ . This information

confirms the coexistence of  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3$ . Fig. 3c are typical EDS elemental mappings of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$  nanosphere, indicating that C, Bi, V, and O are co-existed in  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$  nanospheres.



**Fig. 3.** (a) TEM image and (b) HRTEM image of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ . (c) EDS elemental mapping images of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ .

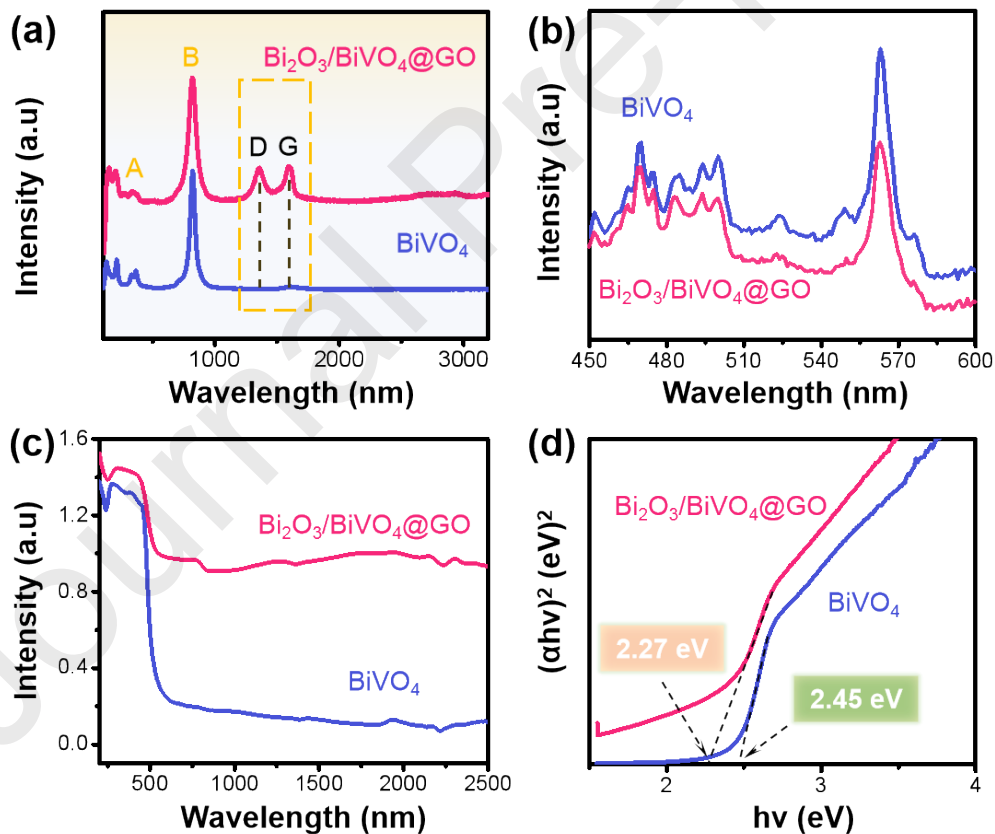
Raman spectroscopy was employed to investigate the bonding and structure of as-prepared pure  $\text{BiVO}_4$  and 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures and the results are shown in **Fig. 4a**. The Raman peak at  $817\text{ cm}^{-1}$  is due to the asymmetric and symmetric V-O stretching modes [27], confirming the presence of as-grown  $\text{BiVO}_4$ . In

addition, compared with pure  $\text{BiVO}_4$ , the spectrum of the  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  displays D and G bands of the GO positioned at 1352 and 1589  $\text{cm}^{-1}$  [28], which confirms that layered GO acts as the skeleton of 2D/3D structure. It can be noted that the intensity ratio of A to B in both  $\text{BiVO}_4$  (0.13) and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  (0.1) was almost the same, demonstrating that the GO coating have not adversely affected the structure of  $\text{BiVO}_4$ .

The separation efficiency and recombination processes of charge carrier can be revealed by photoluminescence (PL) spectroscopy and the results are shown in **Fig. 4b**. During the charge recombination process, lower intensity in PL spectra refers to less photogenerated electron-hole pairs recombining, which can accelerate the photocatalytic reaction. It is worth noting that with GO layer coating, the PL peak intensity of 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructure greatly decreases than that of  $\text{BiVO}_4$ , implying the lower charge recombination for  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructure. This should be attributed to the conjugate structure and excellent carrier mobility of GO which can promote electron and mass transfer and suppress charge carrier recombination [6]. On the other hand, the fabrication of multi-dimensional 2D/3D vdW heterostructures not only enhances the interface contact but also reduces the photogenerated charges transfer distance to improve the separation of charge carriers [23].

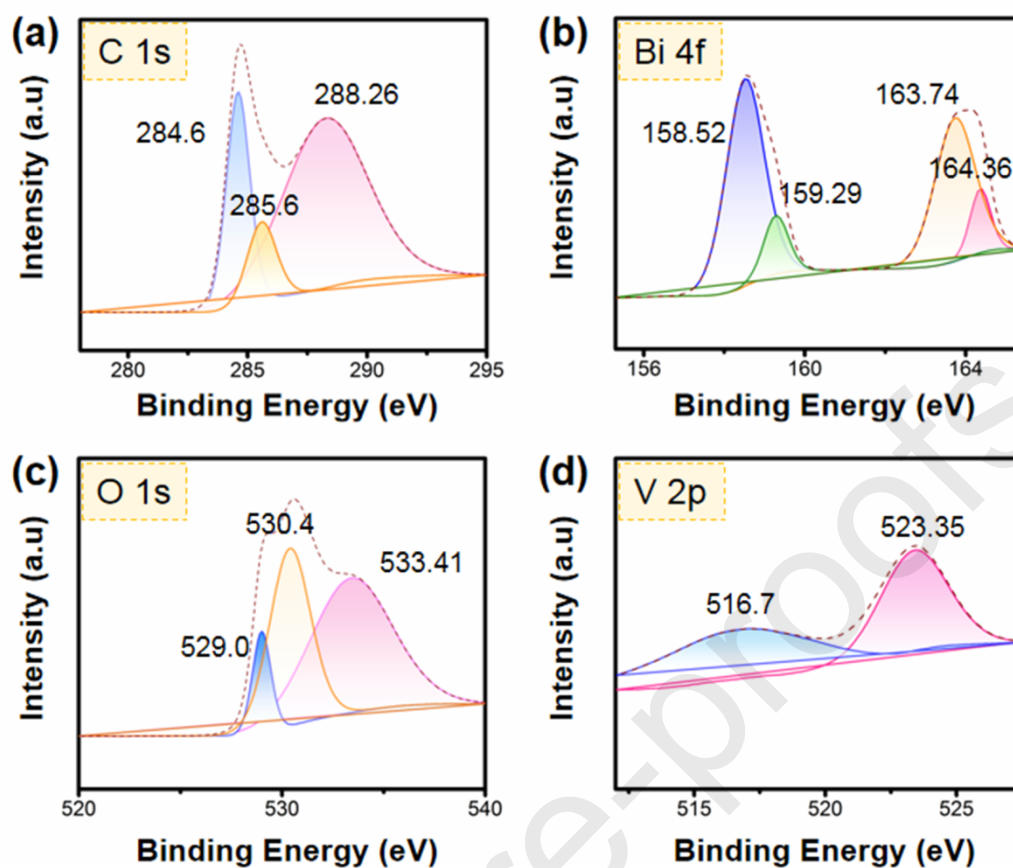
To investigate the optical properties of as-prepared pure  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures, UV-vis absorption spectra were analyzed and the results are depicted in **Fig. 4c**. As can be seen,  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures exhibit evident absorption curves and show different absorption edges at around 450 nm and 530 nm, respectively. In comparison with pure  $\text{BiVO}_4$ , the absorption edge of

$\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures has a distinct red shift, indicating the increased light absorption and enhanced photocatalytic activity. Besides,  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures have the significantly enhanced light absorption in the visible and even near-infrared (NIR) region, which can be attributed to the reduced energy gap and the formation of 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterojunction. Based on above absorption spectra, the band gap ( $E_g$ ) of samples can be calculated [29] and displayed in **Fig. 4d**. The estimated  $E_g$  values for  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  are about 2.45 eV and 2.27 eV, indicating 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  here structures have narrow optical  $E_g$ .



**Fig. 4.** (a) Raman spectra, (b) PL spectra, (c) UV-Vis absorption spectra and (d) bandgap of  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ .

X-ray photoelectron spectroscopy (XPS) was further used to evaluate the surface chemical states of 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures and the results are shown in **Fig. 5**. In the high-resolution C 1s spectrum of GO (**Fig. 5a**), three major peaks located at 284.6, 285.6, and 288.26 eV were attributed to the binding energy of the  $\text{sp}^2$  C=C bonds,  $\text{sp}^3$  C-C bonds and O-C=O bonds, respectively. Two weak peaks at 158.52 eV and 163.74 eV are observed in Bi 4f spectrum (**Fig. 5b**), which correspond to the Bi  $4f_{5/2}$  and Bi  $4f_{7/2}$  binding energies of  $\text{BiVO}_4$ . Meanwhile, the two peaks at 159.29 eV and 164.36 eV are indexed to Bi  $4f_{5/2}$  and Bi  $4f_{7/2}$  in  $\text{Bi}_2\text{O}_3$  (**Fig. 5b**), indicating that Bi element is presented as  $\text{Bi}^{3+}$  state, which suggests  $\text{Bi}_2\text{O}_3$  exists in the heterostructures. As for the O 1s spectrum of **Fig. 5c**, the spectrum can be resolved into three characteristic peaks. The peak appearing at 529 eV can owe to the  $\text{O}^{2-}$  in  $\text{BiVO}_4$ , whereas the main peak at 530.4 eV was associated with C-O bonds in the GO. The other peak at 533.41 eV in O 1s spectrum is attributed to the Bi-O bonds in  $\text{Bi}_2\text{O}_3$ . The V 2p peak splits into two strong peaks at around 516.7 and 523.25 eV, which can be attributed to V  $2p_{3/2}$  and V  $2p_{1/2}$  binding energies of  $\text{BiVO}_4$  (**Fig. 5d**).



**Fig. 5.** XPS spectra of (a) C 1s, (b) Bi 4f, (c) O 1s and (d) V 2p for Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO.

The typical functional groups of BiVO<sub>4</sub> and Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO were investigated by Fourier transform infrared (FTIR) spectra, as shown in **Fig. 6a**. The spectra exhibit band at ~1726 cm<sup>-1</sup> related to the C=O stretching of COOH groups at the edges of GO sheets. The bands at ~3500 cm<sup>-1</sup> and 1630 cm<sup>-1</sup> are related to OH stretching and bending vibrations [30-32]. The vibrational band at ~1400 cm<sup>-1</sup> is attributed to the OH vibrations of H<sub>2</sub>O molecules adsorbed on the sample surface. Furthermore, the band at ~739 cm<sup>-1</sup> ascribes to the asymmetric stretching vibration of the ν<sub>3</sub> (VO<sub>4</sub>) [33]. The band at ~526 cm<sup>-1</sup> is ascribed to the Bi-O stretching pattern of Bi<sub>2</sub>O<sub>3</sub> [34, 35], indicating the existence of Bi<sub>2</sub>O<sub>3</sub> in the 2D/3D heterostructure. Additionally, it is obvious that the

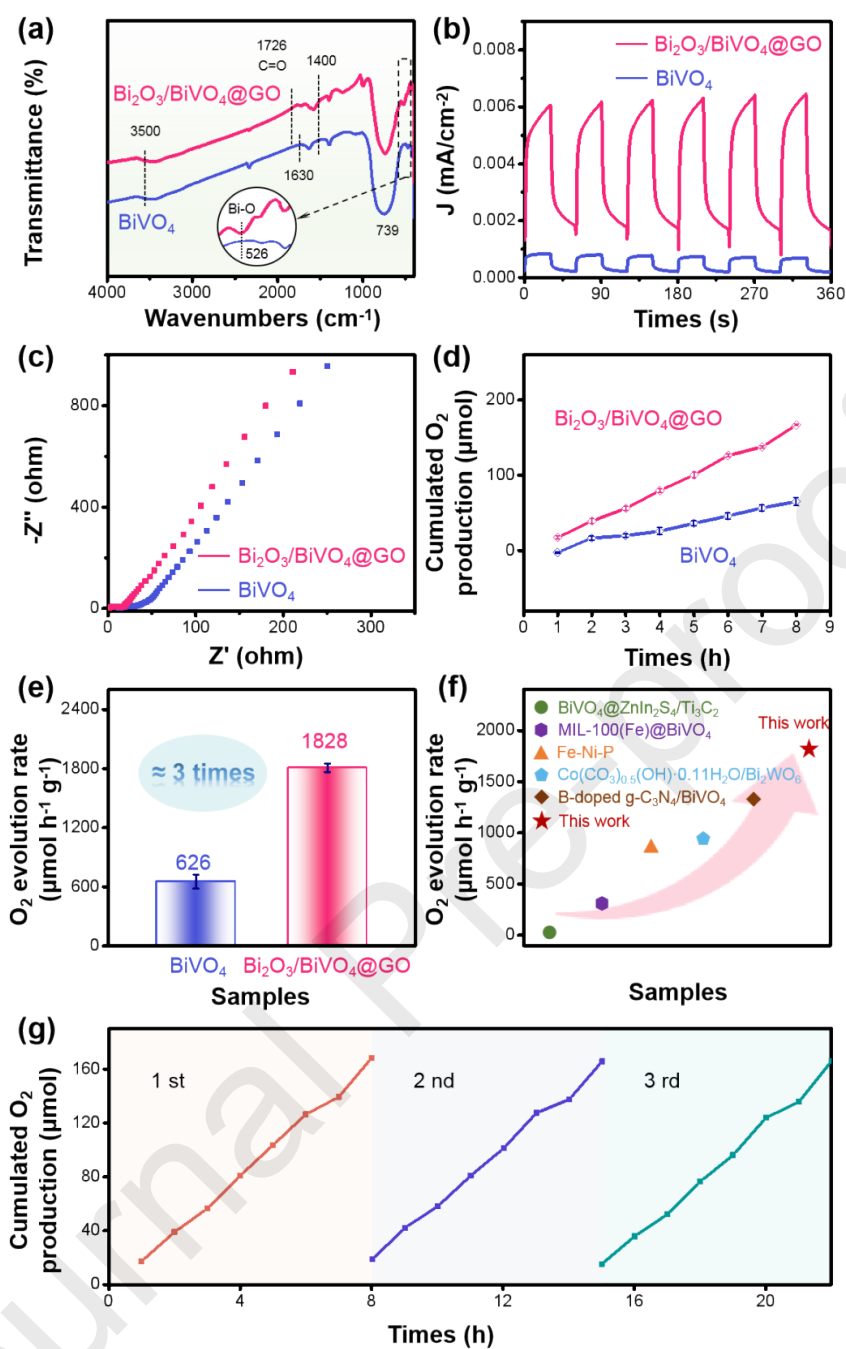
primary bands of  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3$  are included in 2D/3D structure, implying that the  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures are fabricated successfully. Meanwhile, the spectra of 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructure are similar to those of the main peaks of pure  $\text{BiVO}_4$ , demonstrating that the structure of  $\text{BiVO}_4$  remains unchanged wrapped with the GO framework.

In order to further analyze the separation and transfer of photogenerated carriers in 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures, their reaction process was evaluated by photocurrents and electrochemical impedance spectroscopy (EIS). **Fig. 6b** displays the photocurrent responses of  $\text{BiVO}_4$  and 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures. As can be seen, 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures show a much higher photocurrent density at a given potential comparing with that of pure  $\text{BiVO}_4$ , suggesting 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures have much higher transfer and separation efficiency of charge carriers. Moreover, the photocurrent remains stable after six repeated on/off light cycles, as suggested in **Fig. 6b**. As shown in **Fig. 6c**, the comparison of the Nyquist plots exhibits a much smaller arc radius of 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures than that of pure  $\text{BiVO}_4$ , confirming smaller charge transfer resistance of 2D/3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures. Note that the GO loading in the heterostructure constitutes a framework for enhancing the transport of charge carriers while the 2D/3D structure serves to provide active pathway and massive reactive sites, providing a pathway for charge carrier migration rate.

**Fig. 6d** plots the photocatalytic  $\text{O}_2$  evolution results for pure  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures were investigated within 5 h irradiation. Apparently, the pure  $\text{BiVO}_4$  can catalyze the oxidation of water to  $\text{O}_2$  and

the amount of evolved  $O_2$  is 62.58  $\mu\text{mol}$ . In contrast,  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterostructures exhibits higher activity for  $O_2$  evolution than pure  $\text{BiVO}_4$  and gives 168.23  $\mu\text{mol}$  of  $O_2$ , implying the superiority of 2D/3D heterostructures. As shown in **Fig. 6e**, the  $O_2$  evolution rate of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  composite is 1828  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , nearly 3 times higher than that of the pure  $\text{BiVO}_4$ . **Fig. 6f** plots the comparison between our heterostructures and the reported [36-40]. As can be seen, our heterostructures show one of best performance for photocatalytic  $O_2$  evolution activity. This remarkable increase in the  $O_2$  evolution of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  is largely due to the structural coupling of the 2D GO and 3D  $\text{Bi}_2\text{O}_3/\text{BiVO}_4$  heterojunction, allowing short electron transport distance and rapid charge transfer at the interface. Meanwhile, the heterojunction provides massive reactive sites and accelerates charge separation and transfer processes, taking advantage of the high specific surface area and good carrier mobility of GO framework. As shown in **Fig. 6g**, no obvious  $O_2$  evolution rate decrease was observed after three cycles, demonstrating the stability of  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ .

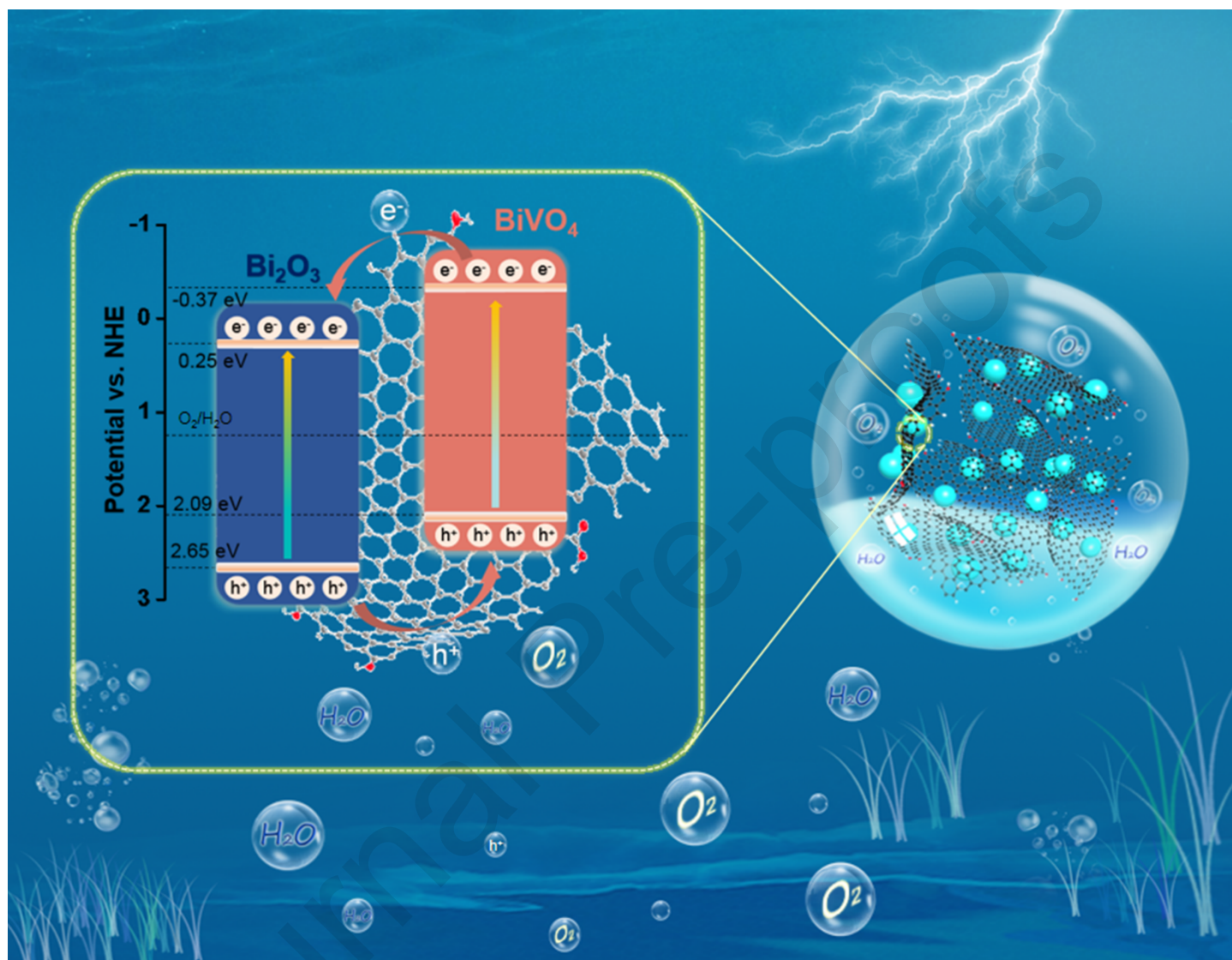




**Fig. 6.** (a) FTIR spectra of  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ . (b) Photocurrent responses and (c) EIS Nyquist plot of  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ . (d) Photocatalytic  $\text{O}_2$  evolution performance and (e)  $\text{O}_2$  evolution rates of  $\text{BiVO}_4$  and  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ . (f) Photocatalytic  $\text{O}_2$  evolution rates of different samples. (g) Cyclic runs for the photocatalytic  $\text{O}_2$  evolution over  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$ .

According to the above analysis and previous reports [23, 41], **Fig. 7** schematically illustrates the possible photocatalytic mechanism for the enhanced  $O_2$  evolution activity of  $Bi_2O_3/BiVO_4@GO$ . The valence band of  $BiVO_4$  is favorable for  $O_2$  evolution [42]. Under the solar-simulated light irradiation, both  $BiVO_4$  and  $Bi_2O_3$  are excited and produce electron-hole pairs. The electrons on the valence band (VB) can be excited to the conduction band (CB) by the high energy photon, and the equal photogenerated holes ( $h^+$ ) are generated. Generally, the photoinduced electrons ( $e^-$ ) would migrate to the less negative CB, while the holes transfer to the less positive VB [43]. Therefore, due to the more positive CB and VB levels of  $Bi_2O_3$ ,  $e^-$  will shift from  $BiVO_4$  to  $Bi_2O_3$  and  $h^+$  migrates from  $Bi_2O_3$  to  $BiVO_4$ , which can enhance charge separation and improve photocatalytic activity [44]. The  $h^+$  transferred to  $BiVO_4$  can easily oxidize  $H_2O$  to  $O_2$ . The interaction and proper energy band alignment between  $Bi_2O_3$  and  $BiVO_4$  is beneficial to promoting charge migration and suppressing photogenerated carrier recombination [45]. GO provides massive reactive sites and increases the interfacial contact with catalysts. In addition, the nanochannels provided by GO layers are beneficial to shortening migration distance and accelerating electrons transfer in the composite catalysts. GO plays the role of an electron transport mediator that accelerates the interfacial electron-transfer process from  $BiVO_4$  to  $Bi_2O_3$ , strongly inhibiting the recombination of charge carriers and thus improving the photocatalytic activity. Moreover, GO leads to high specific surface area and provides massive reactive sites for  $h^+$  oxidizing  $H_2O$  to  $O_2$  [32, 46]. The improved photocatalytic performance is associated with the enhanced electronic conductivity caused by GO, and the synergetic effects between GO and  $BiVO_4$  and  $Bi_2O_3$  which reduce the recombination of photoinduced charge carriers. In this way, the

heterostructure of type-II is constructed and promotes the charge separation and transfer, in consistence with the results of PL, photocurrent and EIS, which is beneficial for enhanced  $O_2$  evolution activity.



**Fig. 7.** The possible photocatalytic mechanism for the enhanced  $O_2$  evolution activity of  $Bi_2O_3/BiVO_4@GO$ .

#### 4. Conclusion

In conclusion, we use a facile self-assembly technique to fabricate 2D/3D  $Bi_2O_3/BiVO_4@GO$  vdW heterostructures, which exhibits an enhanced  $O_2$  evolution rate of  $1828 \mu\text{mol h}^{-1} \text{g}^{-1}$ , nearly 3 times than that of

pure BiVO<sub>4</sub>. The enhanced photocatalytic activity may originate from the 2D/3D heterojunction with van der Waals interactions. The heterostructures composed of 2D GO and 3D Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub> is expected to play a significant role in promoting electron transfer and providing massive reactive sites. Hence, fabricating 2D/3D vdW heterostructures will create more possibilities in structural design of 2D materials and pave new channels to strengthen the connection between 2D and 3D materials.

### **Author contributions**

Yaxin Bi: Investigation, Data curation, Writing - original draft. Yanling Yang: Conceptualization, Methodology, Writing - review & editing. Xiaolei Shi: Investigation, Writing - review & editing. Lei Feng: Investigation. Xiaojiang Hou: Investigation. Xiaohui Ye: Investigation. Li Zhang: Investigation. Guoquan Suo: Investigation. Jingeng Chen: Investigation. Zhi-Gang Chen: Conceptualization, Methodology, Visualization, Writing - review & editing.

### **Competing interests**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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**The captions of Figures and Tables**

**Fig. 1.** XRD patterns of GO, BiVO<sub>4</sub> and Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO.

**Fig. 2.** (a-c) SEM images and (d) schematic illustration of Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO.

**Fig. 3.** (a) TEM image and (b) HRTEM image of Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO. (c) EDS elemental mapping images of Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO.

**Fig. 4.** (a) Raman spectra, (b) PL spectra, (c) UV-Vis absorption spectra and (d) bandgap of BiVO<sub>4</sub> and Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO.

**Fig. 5.** XPS spectra of (a) C 1s, (b) Bi 4f, (c) O 1s and (d) V 2p for Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO.

**Fig. 6.** (a) FTIR spectra of BiVO<sub>4</sub> and Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO. (b) Photocurrent responses and (c) EIS Nyquist plot of BiVO<sub>4</sub> and Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO. (d) Photocatalytic O<sub>2</sub> evolution performance and (e) O<sub>2</sub> evolution rates of BiVO<sub>4</sub> and Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO. (f) Photocatalytic O<sub>2</sub> evolution rates of different samples. (g) Cyclic runs for the photocatalytic O<sub>2</sub> evolution over Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO.

**Fig. 7.** The possible photocatalytic mechanism for the enhanced O<sub>2</sub> evolution activity of Bi<sub>2</sub>O<sub>3</sub>/BiVO<sub>4</sub>@GO.

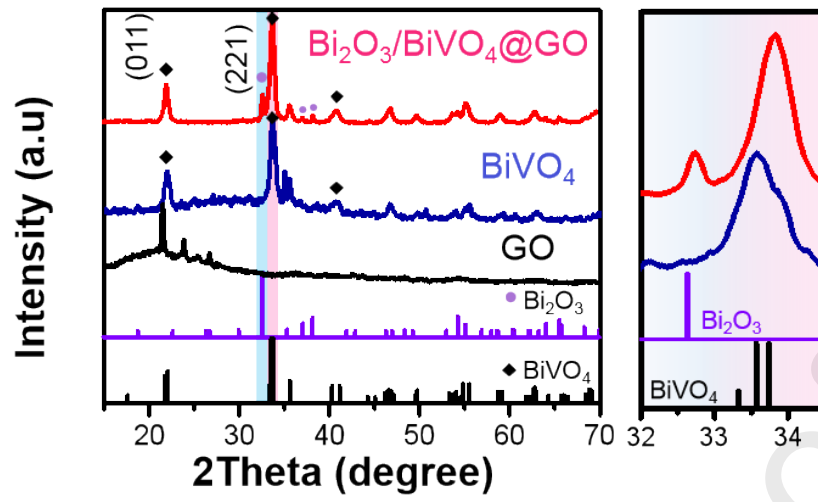


Fig. 1

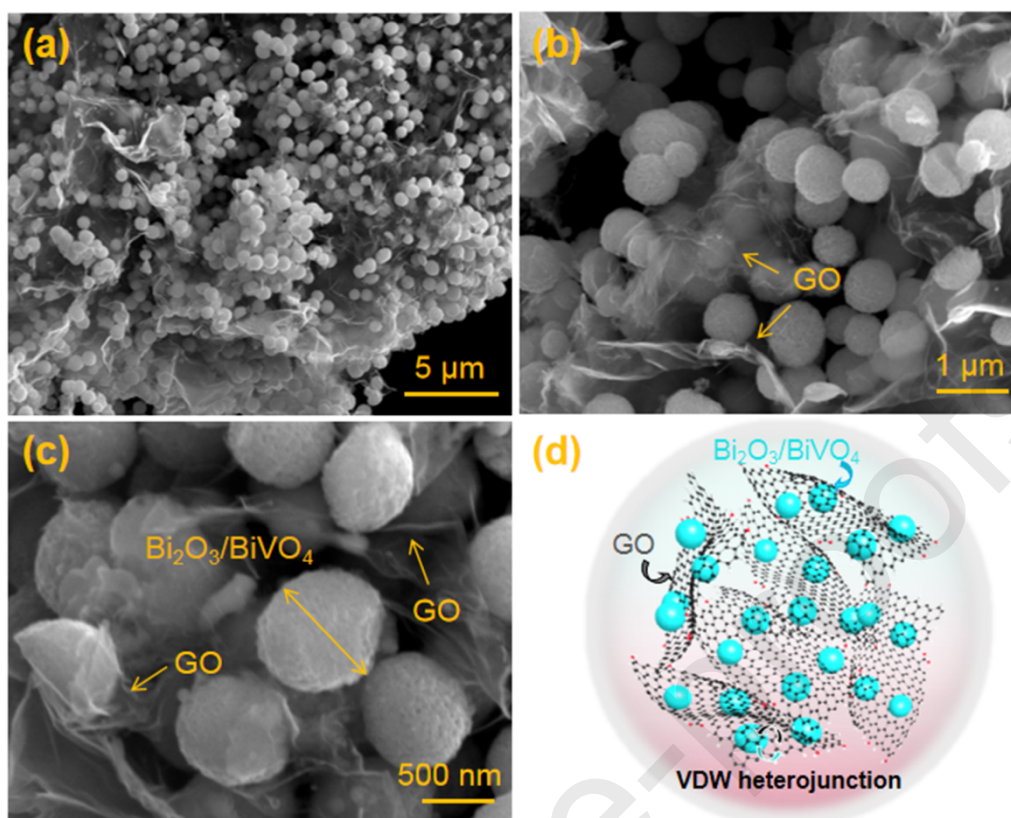


Fig. 2

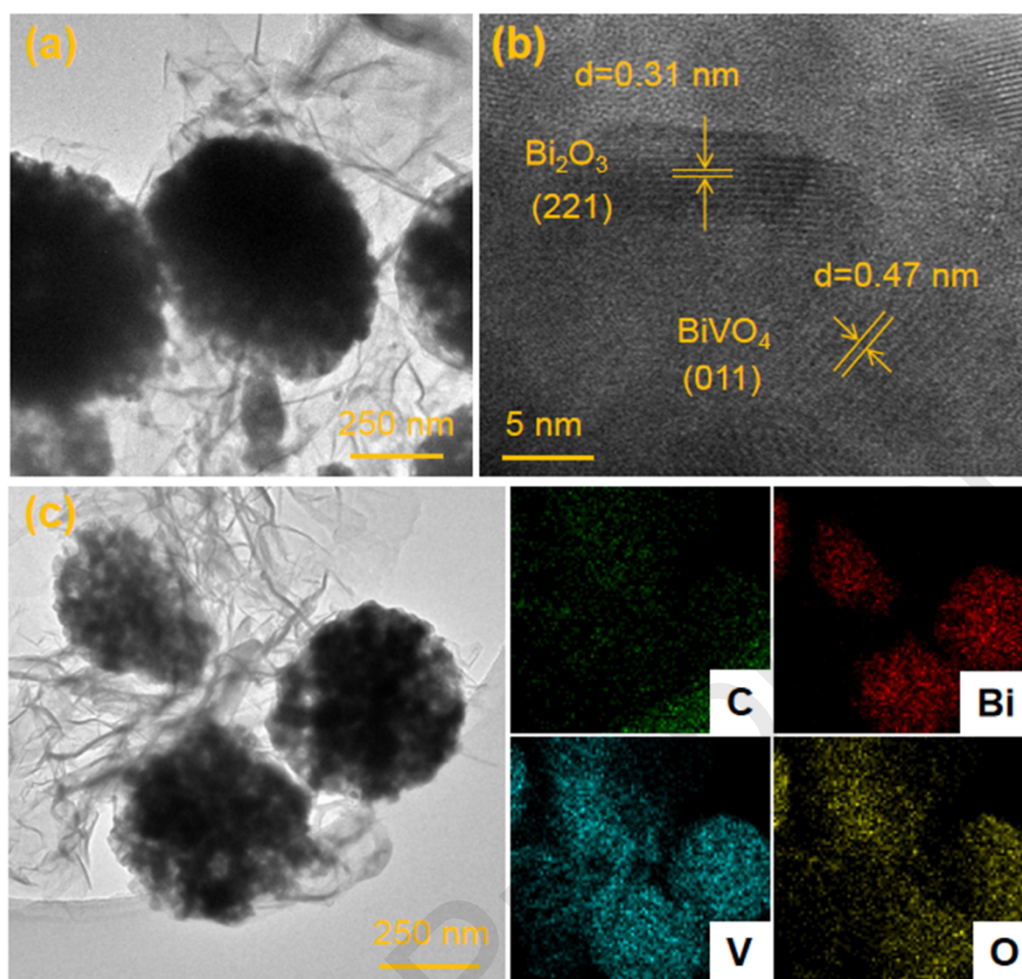


Fig. 3

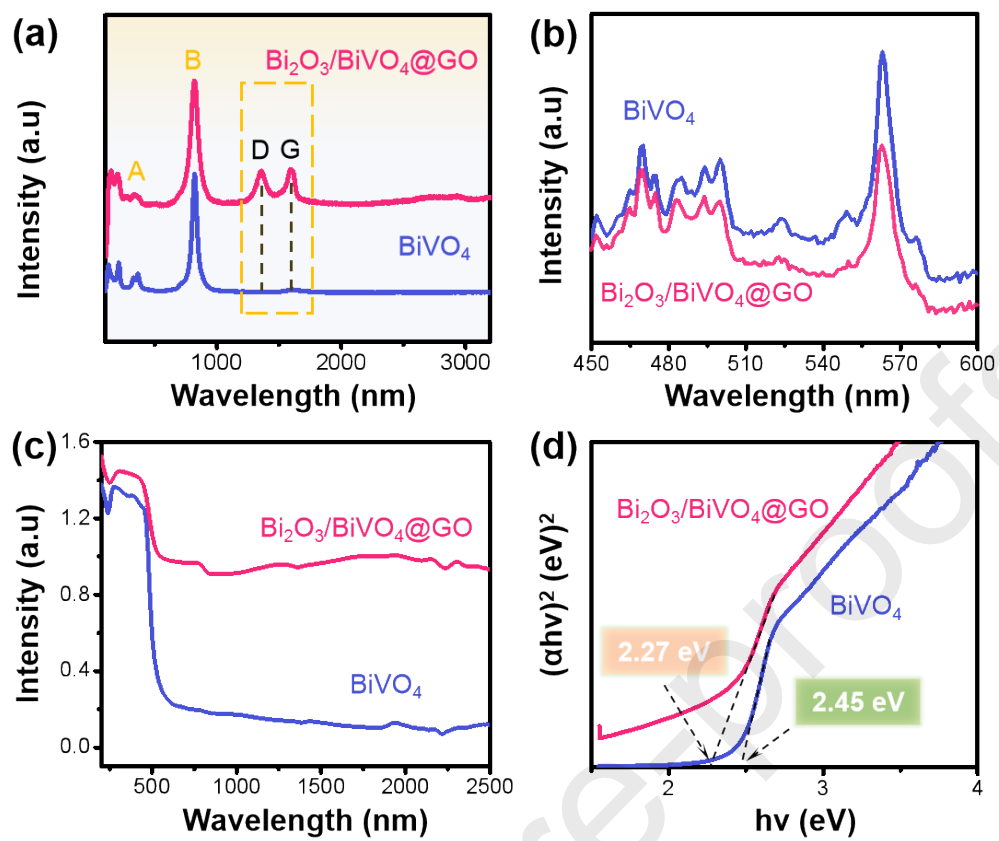


Fig. 4

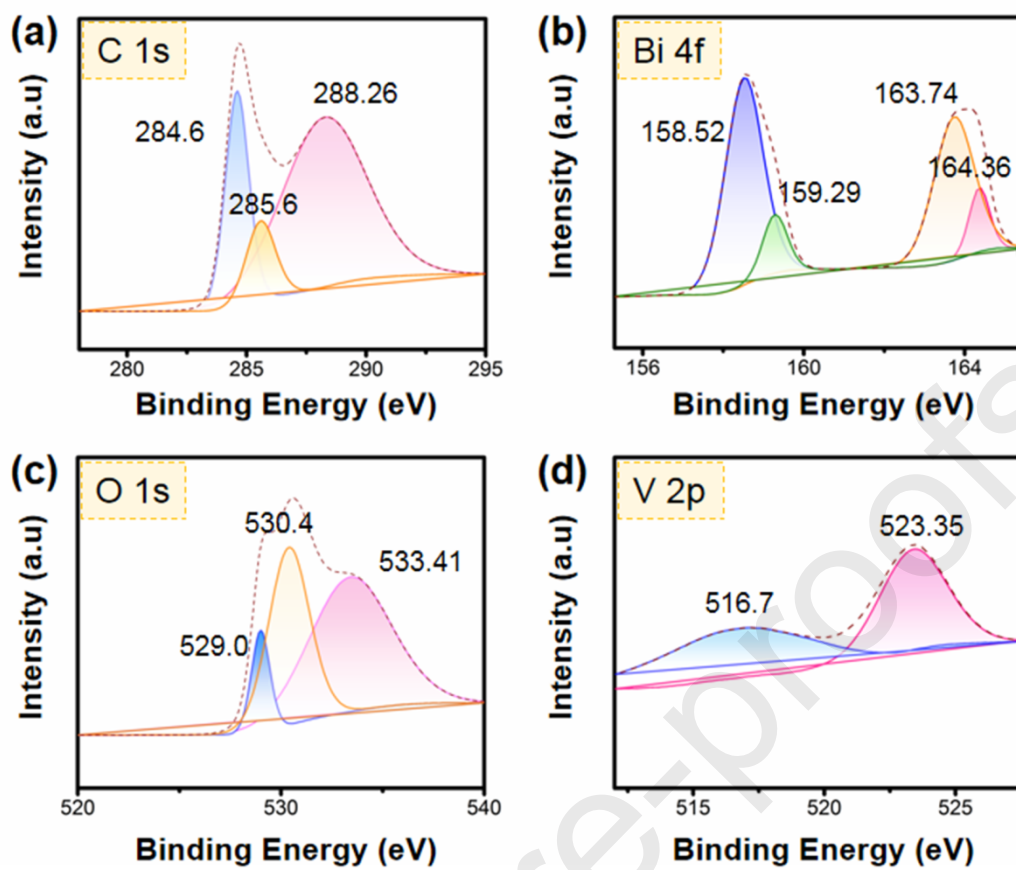


Fig. 5



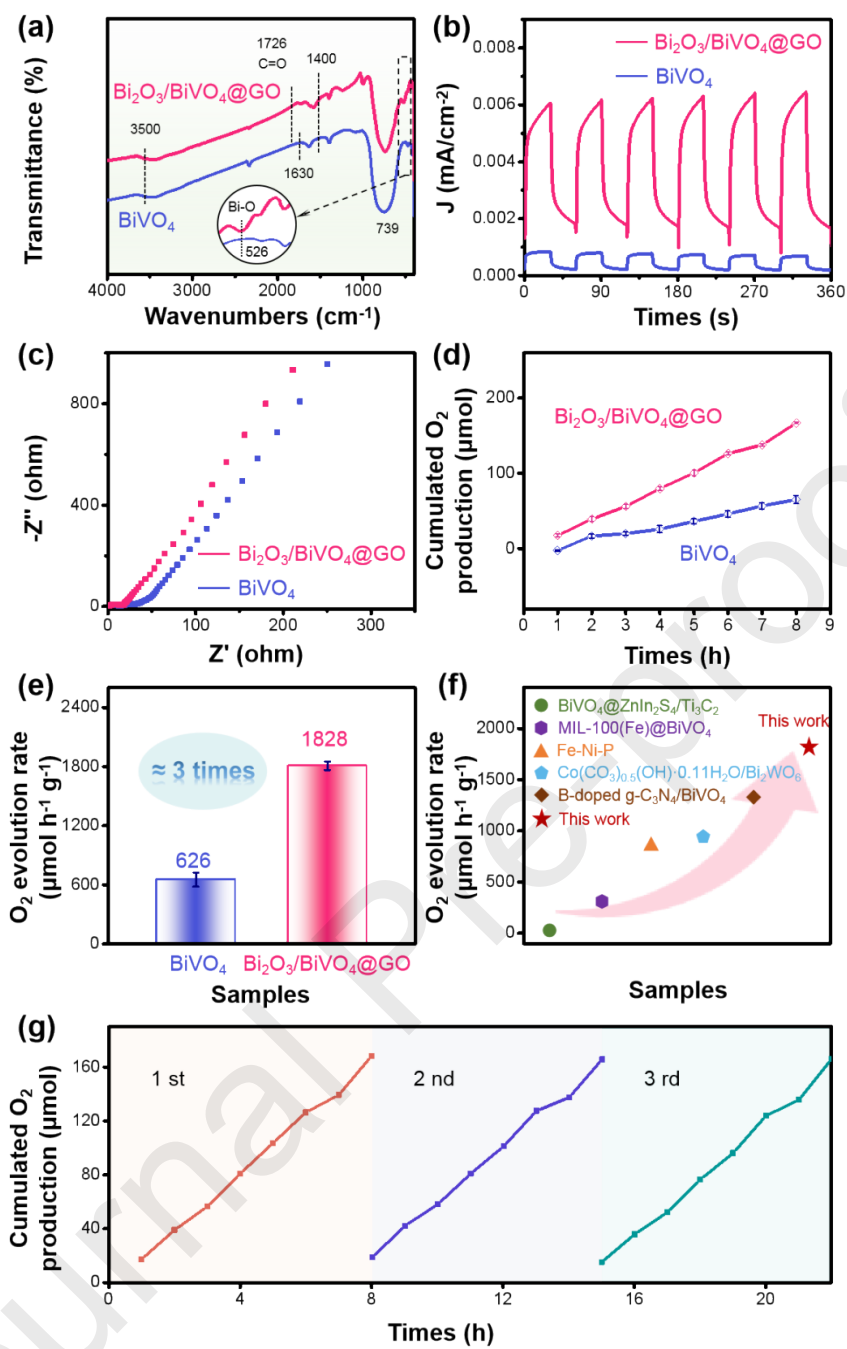


Fig. 6

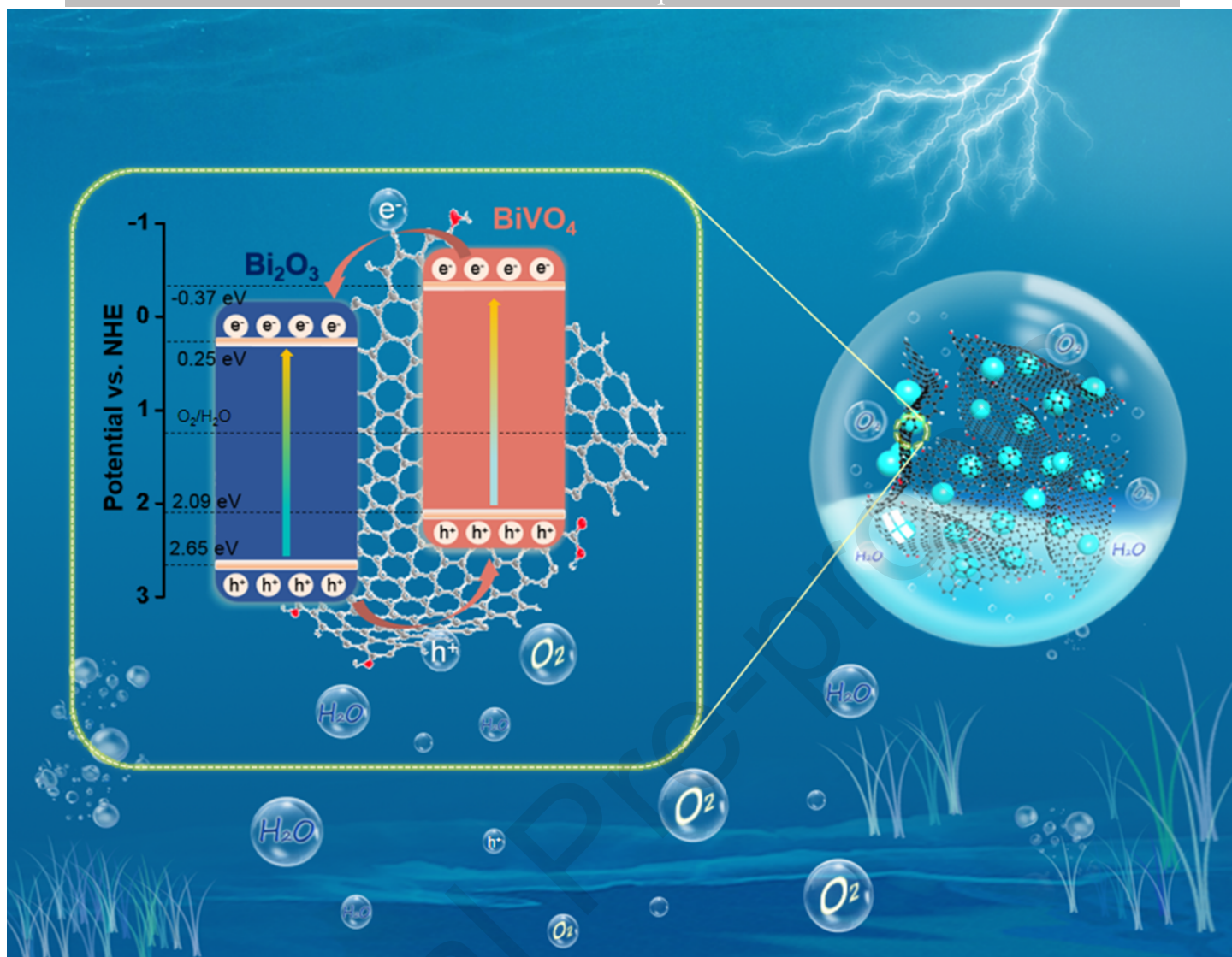
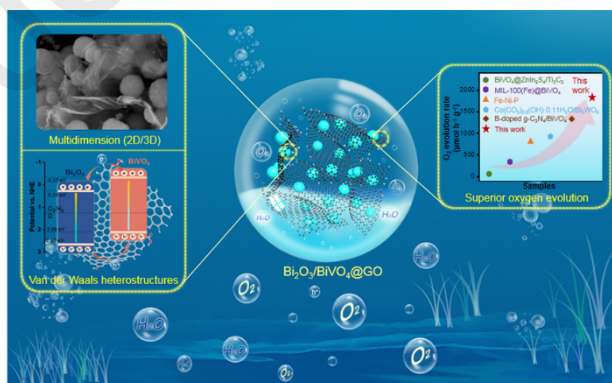


Fig. 7

## TOC



**Highlight**

- A multi-dimensional (2D/3D)  $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  is designed by self-assembly strategy;
- $\text{Bi}_2\text{O}_3/\text{BiVO}_4@\text{GO}$  heterojunction is self-assembled by van der Waals interactions;
- The vdW heterostructure exhibits superior  $\text{O}_2$  evolution of  $1828 \mu\text{mol h}^{-1} \text{g}^{-1}$ .
- The strategy paves new avenue to fabricate multi-dimensional vdW heterostructures.

**CRedit author statement**

**Yaxin Bi**: Investigation, Data curation, Writing - original draft. **Yanling Yang**: Conceptualization, Methodology, Writing - review & editing. **Xiao-Lei Shi**: Investigation, Writing - review & editing. **Lei Feng**: Investigation. **Xiaojiang Hou**: Investigation. **Xiaohui Ye**: Investigation. **Li Zhang**: Investigation. **Guoquan Suo**: Investigation. **Jingeng Chen**: Investigation. **Zhi-Gang Chen**: Conceptualization, Methodology, Visualization, Writing - review & editing.

**Declaration of interests**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

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